

Table 1.2. The six crystal systems.

| System | Latuices | Coordinates | Examples |
| :---: | :---: | :---: | :---: |
| Cubic (isometric) | 3 | $\begin{aligned} & a=b=c \\ & x=\beta=y=90 \end{aligned}$ | $\begin{aligned} & x-\mathrm{Fe} \\ & \mathrm{NaCl} \end{aligned}$ |
| Hexagonal Hexagonal subsystem Rhombohedral subsystem (trigonal) | 1 | $\begin{aligned} & a=b \neq c \\ & \alpha=\beta=90^{\circ} \end{aligned}$ | Graphite $\mathrm{MoS}_{2}$ $i=120^{\circ}$ |
| Tetragonal | 2 | $\begin{aligned} & a=b \neq c \\ & x=\beta=\gamma=90^{\circ} \end{aligned}$ | White Sn PtS |
| Rhombohedral (trigonal) subsystem ${ }^{3}$ | 1 | $\begin{aligned} & a=b=c \\ & x=\beta=\eta \neq 90^{\circ}<120^{\circ} \end{aligned}$ | Calcite $\mathrm{K}_{2} \mathrm{Cr}_{2} \mathrm{O}_{4}$ |
| Orthorhombic | 4 | $\begin{aligned} & a \neq b \neq c \\ & x=\beta=\gamma=90^{\circ} \end{aligned}$ | Rhombic S $x-N p$ |
| Monoclinic | 2 | $\begin{aligned} & a \neq b \neq c \\ & x=y=90^{\circ} \neq \beta \end{aligned}$ | Monoclinic S $\mathrm{KICl}_{2}$ |
| Triclinic | 1 | $\begin{aligned} & a \neq b \neq c \\ & x \neq \beta \neq \gamma \end{aligned}$ | $\begin{aligned} & \mathrm{K}_{2} \mathrm{Cr}_{2} \mathrm{O}_{7} \\ & \mathrm{Cu}\left(\mathrm{ClO}_{4}\right)_{2} \cdot 6 \mathrm{H}_{2} \mathrm{O} \end{aligned}$ |

- The rhombohedral subsystem is sometimes categorized as an independent system; however, American crystallographers treit it as at hexigonal subgroup. since the rhombohedral cell can also be described with hexagonal coordinates.


Figure 1.6.
The 14 Bravais lattices with the letters designating P (primitive), I (body-centered), F centered), C (end-centered), and R (rhombohedral). ${ }^{\ddagger}$ Note that a rhomb, which is 1 . hexagonal unit cell (see Fig. 1.20), contains all the symmetry elements of this structure.

## 1. Symmetry



Figure 1.4.
Major planes and axes in the cubic system: (a) the three symmetry planes parallel to the cube faces, (b) the six diagonal mirror planes, (c) the three fourfold rotation axes $\bigcirc$, (d) the four threefold axes $\Delta$, and (e) the six twofold axes of rotation $O$.


(111)


Figure 1.11.
Prominent planes of the cubic system.


Figure 1.13.
The tetrahedrally coordinated interstitial atom in the bec structure.
Figure 1.12.
The body-centered cubic structure showing the cight nearesi neighbors and six next-n neighboring atoms.


Figure 1.14.
The octahedral interstitial sitc in the boc erystal structure

Table 13. Bcc elements.

| Temperature |  | $a_{0}(\AA)^{6}$ | Temperature |  |  |  |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Ba |  | 5.025 | 1002 | $\beta$-Sm | 1190- $T_{\text {m }}$ | - | 1345 |
| $\beta$-Ca | $720-T_{\text {m }}$ | 5.38 (773) | 1112 | $\gamma-\mathrm{Sr}$ | $830-T_{\text {m }}$ | 4.85 (887) | 1041 |
| $\delta$-Ce | 999- $T_{\text {m }}$ | - | 1071 | Ta |  | 3.3058 | 3269 |
| Cr |  | 2.8839 | 2130 | $\beta$-Tb | ${ }^{150-T} \mathrm{~T}_{\mathrm{m}}$ | - | 1630 |
| Cs |  | 6.067 (78) | 301.5 | $\beta-\mathrm{Th}$ | $1636-T_{\text {m }}$ | 4.11 (1723) | 2028 |
| $\beta$-Dy | $1657-T_{\text {m }}$ | - | 1682 | $\beta-\mathrm{Ti}$ | $1155-T_{m}$ | 3.3065 (1173) | 1943 |
| Eu |  | 4.606 | 1090 | $\beta-\mathrm{Tl}$ | 507-T ${ }_{\text {m }}$ | 3.882 | 577 |
| $\alpha-\mathrm{Fe}$ | $<1183$ | 2.8665 |  | $\gamma-\mathrm{U}$ | 1048- $T_{\text {m }}$ | 3.474 | 1405 |
| $\delta-\mathrm{Fe}$ | $1667-T_{\text {mi }}$ | 2.94 (1698) | 1809 | V |  | 3.0240 | 2175 |
| $\beta$-Gd | $1533-T_{\text {m }}$ | - | [595 | W |  | 3.16469 | 3653 |
| $\beta$-Hf | 2013- $T_{\text {m }}$ | - | 2227 | $\beta-\mathrm{Yb}$ | 1033- $T_{\text {m }}$ | - | 1097 |
| $\beta$-Ho | $1701-T_{m}$ | - | 1843 | $\beta-\mathrm{Zr}$ | 1036- $T_{\text {m }}$ | 3.62 (1123) | 2125 |
| K |  | 5.247 (78) | 336.4 |  |  |  |  |
| $\gamma$-La | $1134-T_{\text {m }}$ | - | 1193 |  |  |  |  |
| $\beta$-Li | $80-T_{\text {m }}$ | 3.5093 | 454 |  |  |  |  |
| $\delta$-Mn | $1410-T_{\text {m }}$ | 3.0806 (1407) | 1517 |  |  |  |  |
| Mo |  | 3.1473 | 2890 |  |  |  |  |
| $\beta-\mathrm{Na}$ | 40- $T_{\text {m }}$ | 4.2906 | 371 |  |  |  |  |
| Nb |  | 3.3004 | 2740 |  |  |  |  |
| $\beta-\mathrm{Nd}$ | $1128-T_{m}$ | 4.13 (1156) | 1283 |  |  |  |  |
| $\beta-\mathrm{Pr}$ | $1068-T_{\text {m }}$ | 4.13 (1094) | 1204 |  |  |  |  |
| $\varepsilon$-Pu | 753-Tm | 3.638 (773) | 913 |  |  |  |  |
| Rb |  | 5.605 (78) | 312 |  |  | - | . |
| $\beta$-Sc | 1608- $\mathrm{T}_{\text {ra }}$ | - | 1812 |  |  | . |  |

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gure 1.20. $\qquad$
te axes and indices of some of the prominent planes in the hexagnal system.


Figure 1.16.
The hap unit cell as constructed from equidimensional spheres and the lattice upon which this structure is based.

Table 1.5 Hcp elements.

| Element | Temperature <br> Range (K) | $a_{0}(\AA)^{\text {a }}$ | $c_{0}(\dot{A})$ | $T_{\text {m }}(\mathrm{K})$ |
| :---: | :---: | :---: | :---: | :---: |
| Be |  | 2.2866 | 3.5833 | 1527 |
| Cd |  | 2.97887 | 5.61765 | 594 |
| $\beta-\mathrm{Ce}$ | 125-350 (dhep) ${ }^{\text {b }}$ | 3.65 | 5.96 |  |
| $x-\mathrm{Co}$ | $<700$ | 2.5071 | 4.0686 (293) |  |
| $\alpha-D y$ | $<1657$ | 3.5903 | 5.6475 (293) |  |
| Er |  | 3.5588 | 5.5874 | 1795 |
| $\alpha-\mathrm{Gd}$ | $<1533$ | 3.6360 | 5.7026 (293) |  |
| $\alpha-\mathrm{Hf}$ | $<2013$ | 3.1967 | 5.578 (299) |  |
| $x-\mathrm{Ho}$ | $<1701$ | 3.5773 | 5.6158 (293) |  |
| $\alpha$-La | $<550$ (dhcp) | 3.75 | 6.07 |  |
| $\alpha-\mathrm{Li}$ | <80 | 3.111 | 5.093 (78) |  |
| Lu |  | 3.5031 | 5.5509 |  |
| Mg |  | 3.20927 | 5.21033 |  |
| $\alpha-\mathrm{Na}$ | $<40$ | 3.767 | 6.154 (5) |  |
| $\alpha-N d$ | $<1128$ (dhcp) | 3.657 | 5.902 |  |
| Os |  | 2.7352 | 4.3190 | $3300)$ |
| $\alpha-\mathrm{Pr}_{r}$ | $<1068$ (dhep | 3.669 | 5.920 |  |
| Re |  | 2.7608 | 4.4582 | 3453 |
| Ru |  | 2.70389 | 4.28168 | 2523 |
| $\alpha$-Sc | $<1608$ | 3.3090 | 5.2733 (293) |  |
| $\alpha-\mathrm{Tb}$ | $<1560$ | 3.6010 | 5.6930 (293) |  |
| Tc | . | 2.735 | 4.388 | 2473 |
| $\alpha-\mathrm{Ti}$ | $<1155$ | 2.950 | 4.686 (298) |  |
| $\alpha-\mathrm{Tl}$ | < 507 | 3.456 | 5.525 |  |
| Tm |  | 3.5375 | 5.5546 (293) | 1818 |
| $\alpha-Y$ | $<1752$ | 3.6474 | 5.7306 (293) |  |

- All latlice parameters are given for 298 K unless otherwise noted in parentheses.



Figure 1.18.
The solid circle is the octahedrally coordinated interstice at the center of a face-centercd cubic cell.


Figure 1.19.
Illustrating the eight tetrahedral interstitial positions in the fec unit cell.

| Element | Temperature <br> Range (K) | $a_{0}(\AA)^{\prime}$ | $T_{\text {m }}(\mathrm{K})$ |
| :---: | :---: | :---: | :---: |
| Ac |  | 5.31 I | 1323 |
| Ag |  | 4.0862 | 1234 |
| AI |  | 4.04958 | 933 |
| Au |  | 4.07825 | 1337 |
| $\alpha-\mathrm{Ca}$ | $<720$ | 5.576 |  |
| $\alpha-\mathrm{Ce}$ | $<125$ | 4.85 (77) |  |
| $\gamma-\mathrm{Ce}$ | 160-999 | 5.1601 |  |
| $\beta$-Co | $700-T_{\text {m }}$ | 3.548 | 1768 |
| Cu |  | 3.61496 | 1356 |
| $\gamma-\mathrm{Fe}$ | 1184-1665 | 3.5910 (320) |  |
| Ir |  | 3.8394 (321) | 2716 |
| $\beta$-La | 550-1134 | 5.296 |  |
| $\gamma-\mathrm{Mn}$ | 1360-1410 | 3.52 |  |
| Ni |  | 3.52387 | 1726 |
| Pb |  | 4.9505 | 601 |
| Pd |  | 3.8896 | 1825 |
| Pt |  | 3.9231 | 2042 |
| $\delta$-Pu | 480-588 | 4.6370 |  |
| Rh |  | 3.8031 | 2233 |
| $\alpha-\mathrm{Sr}$ | $<830$ | 6.0847 |  |
| $\alpha-\mathrm{Th}$ | $<1636$ | 5.0843 |  |
| $\alpha-\mathrm{Yb}$ | $<1033$ | 5.4862 |  |



Figure 1.21a.
The diamond cubic unit cell presents a face-centered cubic exterior.
(b)


Figure 1.21b.
The $s p^{3}$ hybridization leads to tetrahedral bonding with a bond angle of $109^{\circ} 28^{\prime}$.


Figure 1.22.
The hexagonal crystal structure of graphitc.



Figure 1.26.
The unit cell of rhombic sulfur.


Figure 1.30.
(a) The orthorhombic unit cell of $\mathrm{I}_{2}$ projected on the $x y$ plane, the numbers denoti distance in and above plane of the page. (b) The atoms drawn with correct van radii to illustrate the close packing. (After Ref. 3.)


[^0]:    - Where no temperature range is given, the element has only one crystal structure
    - All lattice parameters are given for $7 \boldsymbol{R}$ K unlese nthmonina man... :- ........

