The “wavy” strategy of Fig. 4A can accommodate only a relatively modest range of applied strains (i.e., up to a few percent, for the designs reported here). A path to displays with high levels of stretchability uses non-co-planar mesh designs adapted from schemes reported for integrated circuits (13). Figure 4B presents optical micrographs of such a system, composed of a 16 by 16 square array of ILEDs bonded to a PDMS substrate and interconnected by electrodes supported by arc-shaped bridges, with a fraction of the pixels turned on (overall yield >80%) (see SOM fig. S15 for details). The shapes of these bridges change in response to deformations of the display, in a way that isolates the ILEDs from any significant strains (figs. S16 and S17). In particular, calculation shows that for strains of 24%, as defined by the change in separation between inner edges of adjacent device islands, the maximum strain in the ILED and quantum well are only 0.17 and 0.026%, respectively. The computed change in emission wavelength is less than ~0.3 nm (see SOM for details). Figure 4C provides optical micrographs of four pixels in this display, in their off and on states, with (top) and without (bottom) external illumination, respectively, in compressed and stretched configurations. The images show the expected reduction in the heights of the arc-shaped bridges that lie in the direction of the applied tensile force (i.e., along the interconnects that run from lower left to upper right), together with an increase in the heights of the bridges in the orthogonal direction, due to the Poisson effect. This mechanical response is fully elastic—the bending-induced strains in the interconnects are small, the strains in the ILEDs are negligible, and the strain in the PDMS is well within its linear response regime. The data in Fig. 4, D and E, are consistent with these mechanics, as are the associated mechanics calculations. In particular, the current-voltage characteristics of a typical device do not change in a measurable way for applied strains up to ~22%, and we observe no degradation on cycling up to a few hundred times (500 times). Recent work demonstrates the use of smaller collections of large, conventional ILEDs in deformable devices that use different designs (24, 25).

The schemes reported here for creating thin, small inorganic LEDs and for integrating them into display and lighting devices create design options that are unavailable with conventional procedures. The planar processing approaches for interconnect resemble those that are now used for organic devices and, for example, large-area electronics for liquid crystal displays, thereby conferring onto inorganic LED technologies many of the associated practical advantages. In large-area, high-pixel count systems (e.g., 1 million pixels per square meter), the ability to use LEDs with sizes much smaller than those of the individual pixels is critically important to achieve efficient utilization of the epitaxial semiconductor material, for reasonable cost. The minimum sizes of devices reported here are limited only by the resolution and registration associated with manual tools for photolithography.

Visualization of Fermi’s Golden Rule
Through Imaging of Light Emission from Atomic Silver Chains

Chi Chen, C. A. Bobisch, W. Ho

Atomic-scale spatial imaging of one-dimensional chains of silver atoms allows Fermi’s golden rule, a fundamental principle governing optical transitions, to be visualized. We used a scanning tunneling microscope (STM) to assemble a silver atom chain on a nickel-aluminum alloy surface. Photon emission was induced with electrons from the tip of the STM. The emission was spatially resolved with subnanometer resolution by changing the tip position along the chain. The number and positions of the emission maxima in the photon images match those of the nodes in the differential conductance images of particle-in-a-box states. This surprising correlation between the emission maxima and nodes in the density of states is a manifestation of Fermi’s golden rule in real space for radiative transitions and provides an understanding of the mechanism of STM-induced light emission.

T he scanning tunneling microscope (STM), which is based on the tunneling effect, has been used to visualize various quantum phenomena in real space, including the quantum corral (1), quantum mirage (2), and particle-in-a-box states (3, 4). All of these demonstrations involved the localization of the electron density of states in confined nanostructures. Light emission from the STM junction reveals a different kind of quantum phenomenon that involves the optical transitions and inelastic electron tunneling (IET) processes in single molecules (5, 6) and nanostructures (7). Furthermore, photon intensity imaging with atomic resolution has been demonstrated (8–10). The spatial resolution in these optical experiments originates from the precision of the STM in injecting electrons in a confined space, although the emitted photons are collected in the far field. This atomic-scale optical detection can reveal aspects of the molecules and nanostructures that are hidden when probed with other techniques.

Imaging of STM light emission has not yet been directly correlated with the underlying elec-
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At lower bias voltages (higher biases, the signal from the NiAl substrate surpasses that of Ag10. (D) for 20 s. (compared to the spectrum of the NiAl substrate [gray (A)] at 2.6 V. All spectra were accumulated (at 0.05-V intervals. The horizontal dashed lines (orange) represent the bias voltage for the bottom Ag10 chain (white line). (Right) Topographical image of Ag10 chain (7° rotation from vertical).

The Ag10 topography shown in the figure was taken at a bias of 2.8 V and tunneling current of 1 nA. (B) Schematic representation of photon emissions from two coupled IET processes, involving an electronic transition in resonance with the excitation of plasmon modes in the junction. The radiative decay of plasmons is detected as photons in the far field. In the resonant coupling, the plasmons oscillate at the same frequency as the light emitted in the transition between two electronic states of the Ag10 chain.

Photon emission from the STM junction occurs through radiative decay of localized surface plasmons that are excited by IET processes (5, 12, 13). These junction plasmons decay as broadband emission with multiple peaks. Their spectral shape is primarily determined by the structure of the STM tip apex at the nanometer scale and also by the dielectric properties of both the tip and substrate (14). A schematic illustration of radiative emission in the tunneling process is shown in Fig. 1B.

When electrons tunnel through the junction, both elastic tunneling (ET) and IET are possible. The IET channel can be observed when it couples to the excitation of other physical quantities such as molecular vibrations, substrate phonons, plasmons, or electronic transitions. For example, if the IET couples to a vibrational mode, it can be measured through a change in the conductance (dI/dV) as the voltage across the tunneling gap passes through the value V_m corresponding to the vibrational energy eV_m = hω.
where $e$ is the charge of an electron, $h$ is Planck’s constant divided by $2\pi$, and $\omega$ is angular frequency (15).

In our experimental case, the IET process could be observed because of the coupling with plasmon modes and electronic transitions. Unlike the ET process, which probes the LDOS only at the applied bias voltage, the IET channel involves both the initial and final LDOS of the inelastic process. The IET efficiency is determined by two factors: The first factor involves the radiative transition between the initial and final states in the tunneling process, and the second factor is that the coupling to the plasmon modes in the junction leads to field enhancement of the radiation field. However, no direct experimental evidence could test this IET process, especially when its analysis required the interpretation of local variations in photon intensity imaging (16).

The radiative emission of plasmons can be characterized by “excitation mapping” (Fig. 2, A to C, upper panels), which reveals the spectral structures of the plasmon emission that shift with the bias voltage. From the spectral evolution of the substrate with increasing bias (Fig. 2A, upper panel), two main emission bands were resolved from the tip–NiAl junction; one was in the visible spectrum and the other lies in the near-infrared. The highest photon energy detected was 2.95 eV, even with higher biases. This is strong evidence for the plasmon emission mechanism. If the emission is through different mechanisms (i.e., hot electron and hole recombination), then higher-energy photons should be emitted as the bias is increased.

When the Ag$_{10}$ chain was in the junction, the emission spectral features of the tip–Ag$_{10}$–NiAl junction remained similar to that of tip–NiAl junction; no additional mode was observed (Fig. 2, B and C, bottom panels). This observation is distinct from theoretical predictions of localized plasmon modes of free atomic chains (17). Because the length of Ag$_{10}$ from the topographic image is 3.2 to 3.4 nm, depending on the bias, it may be too confined to support localized plasmon modes (18).

Although no new plasmon mode was found, the spectral shape and intensity were affected by the presence of the Ag$_{10}$ chain. First, the overall intensity was enhanced with Ag$_{10}$ in the junction (~7.5 times stronger at the center) as compared with bare NiAl (comparing Fig. 2B to 2A). Second, the relative intensity of the visible band was enhanced over the near-infrared band, comparing the upper panels of Fig. 2B and 2C with Fig. 2A, especially the mode at $h\nu = 2.2$ eV, where $h$ is Planck’s constant and $\nu$ is frequency. The introduction of Ag$_{10}$ in the junction changed the final state of inelastic electron tunneling from that around the Fermi level of NiAl to states of the Ag$_{10}$ chain. Thus, the bias for IET excitation of the same plasmon modes was up-shifted by about 0.8 V at the center and by 1.1 V toward the end of the chain (Fig. 2, B and C, upper panels), compared to NiAl (Fig. 2A, upper panels). This up-shift in the excitation bias and the invariance of the plasmon modes have also been observed for light emissions from molecular adsorbate on the metal surface (7, 19).

An additional ~0.3 V is required to excite the same plasmon modes at the end as compared to the center of the Ag$_{10}$ chain. This up-shift in excitation energy has its origin in the $dI/dV$ spectra shown in Fig. 2D, where lower-energy states dominate in the central part of Ag$_{10}$, whereas higher-energy states populate preferentially toward the ends of the chain. This variation of LDOS of Ag$_{10}$ also occurs in atomic Au and Pd chains (3, 4, 20), leading to a difference in their spectral intensity at the same bias. For example, as shown in the bottom panels of Fig. 2, B and C, the overall intensity around $h\nu = 2.4$ eV is higher at the center than at the ends for a bias of 3.4 V. If the dashed line in Fig. 2C is shifted up by 0.3 V, then the spectrum becomes similar to that of 3.4 V at the center. Qualitatively, the shift of the excitation bias with chain position follows the same trend as the shift in $dI/dV$ (7). However, the quantitative relation of these two shifts is complex, because the electronic states at the ends of the chain involve a dramatic transition from Ag$_{10}$ to NiAl.

Clear contrasts in emission intensities along Ag$_{10}$ were obtained by plotting a series of spectra for different positions along the chain to yield a 2D intensity contour (Fig. 2E, left panel). Comparison with the corresponding topography (Fig. 2E, right panel) shows that the photon emission is confined within the chain. For lower photon energies ($h\nu < 1.7$ eV), the emission is distributed throughout the chain. For higher photon energies, the emission intensity is seen to spatially narrow toward the center. In addition, the distribution along Ag$_{10}$ is not uniform in intensity and contains emission maxima and minima. Roughly, the number of maxima decreases from three or four to one as photon energy increases. The observed emission pattern and change in the number of maxima suggest a close connection between the optical transition and the particle-in-a-box properties of Ag$_{10}$.

This connection between $dI/dV$ and photon images is revealed in Fig. 3 for decreasing photon energies. The oscillating electron densities for the different final states, $dI/dV (f)$, are shown horizontally in Fig. 3D. The number of maxima and minima (nodes) in the $dI/dV (f)$ images compared from one to another. (D) $dI/dV (f)$ images for the five final states showing zero, one, two, three, and four modes (dark bands) for the ground state and first, second, third, and fourth excited states of the particle in a box. The positions of the emission maxima coincide with the nodes in $dI/dV (f)$ images. (E) Topographic images $Z(f)$ of the Ag$_{10}$ chain at biases corresponding to different final states $V(f)$. The images of this chain yield a length of 2.9 to 3.4 nm, with the variation due to the dependence of the image on bias voltage. Images for (B), (D), and (E) have the same size: 3.9 nm $\times$ 1.95 nm (64 $\times$ 32 pixels), taken at 1-nA tunneling current.

![Fig. 3.](image-url)
increases with the energy of the states [\(eV_{(f)}\)]. The ends of \(Ag_{10}\) appear as two bright lobes due to the enhancement of edge effects during \(dl/dV\) imaging (4). As the bias \(V_{(f)}\) increases, sequential \(dl/dV_{(f)}\) images could be predominantly assigned to the ground, first, second, third, and fourth excited states (\(\Psi_1, \Psi_2, \Psi_3, \Psi_4\), and \(\Psi_5\), respectively) of this 1D particle-in-a-box system. At each chosen bias voltage, the density does not correspond to a pure state and is a weighted sum of nearby states (3). The voltages are chosen to reveal the dominance of one state exhibiting zero, one, two, three, or four nodes in \(dl/dV_{(f)}\). As expected, separations between the nodes become smaller for higher-energy states. When the bias voltage is higher than 2.8 V, \(dl/dV\) is dominated by the NiAl states (Fig. 2D), and its fine structures are smeared out. For example, at 3.45 V (Fig. 3B), the profile of \(Ag_{10}\) is still discernible, but oscillating patterns can no longer be identified.

The role of the spatially oscillating electronic states of \(Ag_{10}\) in the optical transitions is determined from imaging of the photon emission. Photon imaging of different emission energies at a fixed bias of 3.45 V allows the investigation of final-state effects and is shown schematically in Fig. 3A. Not every final state could be probed, because of the limited photon detection window of the charge-coupled device (1.3 to 3.0 eV) and spectral range of the plasmon modes. For 3.45-V bias, final states from \(\Psi_1\) to \(\Psi_4\) can be accessed. With emission energies from 2.85 to 1.55 V, photon images show zero (nearly no signal), one, two, three, and four bright maxima (Fig. 3C). A comparison with the corresponding final-state \(dl/dV_{(f)}\) images (Fig. 3D) reveals that the number of emission maxima is equivalent to the number of nodes. Furthermore, the distributions of emission maxima directly coincide with the positions of nodes in \(dl/dV_{(f)}\) images. The entire emission spectrum was recorded at each pixel of the x-y image. The different images in Fig. 3C correspond to integrated energy slices within each spectrum. These results are consistent for different chain lengths, scanning directions, and biases. We can completely exclude any influence on the photon emission images from feedback or tip movement (scanning directions, and biases). The rapid rise in LDOS outside the chain in Fig. 4A shows two nodes correspond to minima in \(\psi_{(f)}^2\) and extrema in \(\partial\psi_{(f)}/\partial x\) (Fig. 4, D and E). There are, in total, four extrema in \(\psi_{(f)}^2\), two at the nodes and two at the ends. However, photon emission is observed only at the nodes. These results are consistent with Fermi’s golden rule, because the matrix element contains not only the derivative of the final state \(\psi_{(f)}\), but also the overlap with the initial state \(\psi_{(i)}\). The LDOS for the initial state \(\psi_{(i)}\) vanishes at the ends for a particle in a box with infinite walls and are nearly zero for the \(Ag_{10}\) chain (Fig. 4A). The rapid rise in LDOS outside the chain in Fig. 4A is associated with the NiAl substrate states (Fig. 2D). At high bias values, the \(Ag_{10}\) chains no longer maintain their distinct particle-in-a-box characters because of strong mixing with the NiAl states.

The \(Ag_{10}\) chain is adsorbed on the metallic NiAl(110) substrate. Along the [110] direction, NiAl has a band gap that is responsible for the low differential conductance (\(dl/dV\)) up to ~2.0 V bias voltage (Fig. 2D) (22). The troughs on the surface provide a template for constructing the 1D Ag chains. Because of the band gap, the particle-in-a-box states are weakly coupled to the substrate states and remain distinct for states below ~2.0 V. The radiative process is shown schematically in Fig. 1B, where the intensity of the detected light is enhanced, arising from the coupling of the electronic transition for the plasmon modes in the junction. When the coupling is in resonance, light detected in the far field is enhanced. Such a plasmon-enhanced field is quite general and is determined at the atomic scale in the present experiment. Thus, the use of a Ag tip is preferred over other tips such as W that have strongly damped plasmons. The plasmon modes are revealed by the spectra in the bottom panels of Fig. 2, A to C. The intensity of the radiative plasmon decay does not exhibit spatial variations away from surface inhomogeneities.
Homogeneous Distribution of $^{26}$Al in the Solar System from the Mg Isotopic Composition of Chondrules

Johan Villeneuve,* Marc Chaussidon,1 Guy Libourel1,2

The timing of the formation of the first solids in the solar system remains poorly constrained. Micrometer-scale, high-precision magnesium (Mg) isotopic analyses demonstrate that Earth, refractory inclusions, and chondrules from primitive meteorites formed from a reservoir in which short-lived aluminum-$^{26}$ (26Al) and Mg isotopes were homogeneously distributed at $\pm$10%. This level of homogeneity validates the use of $^{26}$Al as a precise chronometer for early solar system events. High-precision chondrule $^{26}$Al isotopes show that several distinct chondrule melting events took place from $\sim$1.2 million years (My) to $\sim$4 My after the first solids condensed from the solar nebula, with peaks between $\sim$1.5 and $\sim$3 My, and that chondrule precursors formed as early as $0.87^{+0.04}_{-0.16}$ My after.

Models of the evolution of the early solar system rely on knowledge of the precise time scales for the physical and chemical processes that occurred in the early accretion disk and led to the formation of calcium-aluminum-rich inclusions (CAIs) and chondrules, which are the building blocks of primitive meteorites (chondrules). Short-lived $^{26}$Al (half-life ($T_{1/2}$) = 0.73 million years (My)) is possibly the most accurate chronometer for the first few million years of solar system history, provided that it was homogeneously distributed in the disk. Previous studies of various meteoritic components showed that $^{26}$Al was widespread in the disk, but its level of homogeneity has never been quantified precisely. From numerous mineral $^{26}$Al isotopes in CAIs, a canonical value of $\sim 5 \times 10^{-5}$ was inferred for the initial $^{26}$Al/$^{27}$Al ratio [hereafter ($^{26}$Al/$^{27}$Al)$_0$] when CAIs crystallized ($t$). This value is slightly lower than 5.23 ($0.13 \times 10^{-3}$), which was determined from the bulk $^{26}$Al isochron of Allende CAIs (2) [used in our work instead of 5.85 ($0.05 \times 10^{-3}$)], which indicates that the crystallization of CAIs followed closely after (50,000 years at most) the condensation of their precursors. The bulk $^{26}$Al isochron precisely defines an initial Mg isotopic composition ($^{26}$Mg*$_0$) for CAI precursors of $\sim 0.04 \pm 0.029$ per mil (‰) (2). If $^{26}$Al and Mg isotopes were homogeneously distributed in the inner solar system, then CAIs $^{26}$Mg*$_0$ can be taken as the initial Mg isotopic composition of the solar system when $^{26}$Al/$^{27}$Al = 5.23 ($0.13 \times 10^{-3}$). Any object that formed later (when $^{26}$Al/$^{27}$Al < 5.23 $\times 10^{-5}$) must be characterized by a more radiogenic $^{26}$Mg*$_0$ value. The increase of $^{26}$Mg*$_0$ is a function of the Al/Mg ratio of the reservoir in which the object, or its precursors, resided before the last melting event. Al-rich and ferromagnesian chondrules from unequilibrated ordinary chondrites (UOCs) and carbonaceous chondrites (CCs) show mineral $^{26}$Al isotopes with ($^{26}$Al/$^{27}$Al)$_0$ $\sim 2 \times 10^{-3}$ (4–12), that is, much lower than that of CAIs. Because the $^{26}$Mg*$_0$ of chondrules has never been precisely determined, the data from chondrules cannot be compared to those of CAIs nor to the theoretical Mg isotope growth curve calculated for the solar nebula for a homogeneous distribution of $^{26}$Al and Mg isotopes.

We developed high-precision analysis of Al and Mg isotopes using the Centre de Recherches Petrographiques et Geochimiques (CRPG)–CNRS Cameca ion microprobe (ims 1270) to determine precisely (even for low Al/Mg ratios).

References and Notes
11. The details of this experiment are available as supporting material on Science Online.
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